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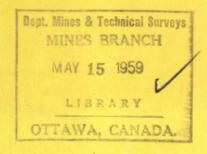
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ON THE OCCURRENCE OF E-CARBIDE IN IRON



by

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ON THE OCCURRENCE OF E-CARBIDE IN IRON*

by F.W.C. Boswell^{**}

ABSTRACT

Quenched and aged alpha iron specimens have been examined by electron diffraction after polishing and etching to leave carbide particles in relief. Specimens aged at 200°C for 2 hours gave patterns typical of hexagonal epsilon iron carbide and those aged at 400°C for 1 hour gave patterns typical of cementite.

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F. W. C. Boswell

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It is well known that \mathcal{E} -iron carbide forms during the first stage of tempering hardened steel (Jack, 1951a; Roberts, Averbach and Cohen, 1953; Lement, Averbach and Cohen, 1954). Single-crystal X-ray work (Roberts et al., 1953; Kurdjumov and Lyssak, 1947, 1949) on this material has shown that the \mathcal{E} -carbide phase exists in metastable equilibrium with martensite containing about 0.25 wt. % carbon and it was concluded (Roberts et al:, 1953) that ε -carbide would not form from martensite containing less than this amount of carbon. It thus appeared unlikely that E-carbide would form directly from supersaturated ferrite. This view was supported by kinetic measurements (Dijkstra, 1949: Wert, 1949) on the precipitation of carbon from quenched iron containing about 0.02 wt. % carbon which gave no indication of the existence of a metastable phase prior to the formation of cementite. It was thus somewhat surprising when Tsou, Nutting and Menter (1952), using electrondiffraction methods, reported the existence of \mathcal{E} -carbide in iron containing 0.026 wt. % carbon. This is a brief report of further electron-diffraction experiments on iron which support the conclusions of Tsou et al.

Both commercial grade (Armco) iron with a nominal carbon content of 0.01 wt. % and high-purity vacuum-melted

iron also containing 0.01 wt. % carbon were examined. The results obtained from the two materials were identical. Electron-diffraction patterns were obtained from specimens immediately after quenching from 720° C after ageing at 200° C for 2 hr. and after ageing at 400°C for 1 hr. The quenched specimens gave only spotty patterns due to the iron itself. The additional face-centred cubic γ -iron phase reported by Tsou et al (1952) was not detected. Those specimens aged at 400°C gave patterns typical of cementite. However, the specimens aged at 200°C for 2 hr. gave rise to a markedly different pattern shown in Fig. 1. The spots are due to the ferrite matrix and measurements of the continuous rings lead to lattice spacings in agreement with those of \mathcal{E} -iron carbide as shown in Table I (Jack, 1951a). It is known from internal friction measurements (Kijkstra, 1949: Wert, 1949) that the carbon precipitation process in iron is essentially complete after ageing at 200°C for 2 hr., hence the most obvious interpretation of the above observations is that a carbide, namely *E*-carbide, is present after this treatment.

Since the occurrence of \mathcal{E} -carbide in iron has not been detected except by electron diffraction experiments, other possible interpretations of the pattern obtained from the specimens aged at 200°C have been considered.

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Table I. Comparison of the interplanar spacings obtained from iron quenched from $720^{\circ}C$ and aged at $200^{\circ}C$ for 2 hr., with those of \mathcal{E} -iron carbide

	Interplanar spacings (kX.)	
<u>kkl</u>	Iron aged at 200 C for 2 hr.	E-Iron carbide*
100 002 101 102 110 103 200) 112) 201)	2.39 Not resolved 2.08 1.59 1.35 1.22 1.17	2.36 2.16 2.07 1.60 1.36 1.23 (1.18 (1.16 (1.14

*Calculated for a = 2.729, c = 4.326 kX. (Jack, 1951a).

The alternative explanations would be to postulate (a) that it was due to an iron nitride phase, or (b) that it was due to two-dimensional diffraction from cementite platelets only a few unit cells in thickness. The first postulate may be ruled out since the nitrogen content of the high-purity iron was only one-tenth that of the carbon content, and thus the intensity of any nitride pattern would be very small compared to that of the carbide. In any case the \mathcal{E} -nitride phase does not occur under these conditions (Jack, 1951b, 1952) and the presence of α^{n} -iron nitride or γ -iron nitride would not account for the observed pattern As regards the possibility of two-dimensional diffraction, this is also unlikely in view of the microscopically observed dimensions of the precipitate particles. Electron-microscopic examination of the high-purity iron specimen aged at

200°C for 2 hr. indicated the presence of precipitate particles which were in the form of thin platelets several thousand Angstrom units in the large dimension and a few hundred Angstrom units thick. It is not possible that cementite platelets of these dimensions could give rise to the patterns observed. It is therefore concluded that the existence of -carbide in iron containing as little as 0.01 wt. % carbon is the most reasonable interpretation of the diffraction patterns from the specimens aged at 200°C.

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